# Mechanochemical Synthesis of LnCoO<sub>3</sub> (Ln: La, Pr, Dy)

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Synthesis of lanthanoid cobaltates,  $LnCoO_3$  (Ln: La, Pr, Dy), via the mechanochemically prepared complex (oxy-)hydroxides has been studied. The hydroxides are prepared by grinding  $Co(OH)_2$  or  $Co_3O_4$  with  $Ln_2O_3$  (Ln: La, Dy) or  $Pr_6O_{11}$  in acetone containing a very small amount of  $H_2O$  in  $LnCoO_3 \cdot 4H_2O$  composition. The starting hydroxides act as the reactive grinding aids to each other.  $La_2O_3$  and  $Co(OH)_2$  form the complex hydroxide,  $LaCo(OH)_5$ , that is a sort of inorganic polymers having La-O-Co bond. The hydration affinity of trivalent  $Co^{3+}$  ions is not high enough to form complex hydroxide,  $LaCo(OH)_6$ . Then, the addition of  $H_2O_2$  to the grinding liquid results in the formation of oxyhydroxide,  $LaCoO(OH)_4$ . These (oxy-)hydroxides are directly converted to pseudo-tetragonal  $LaCoO_3$  at  $600^\circ$ C and rhombohedral  $LaCoO_3$  above  $800^\circ$ C without forming any by-product. The redox interaction between  $Co(OH)_2$  and  $Pr_6O_{11}$ , which shows the poor reactivity with Fe<sub>2</sub>O<sub>3</sub>, enables the formation of  $PrCoO(OH)_4$ . Contrary, the reaction with  $Dy_2O_3$  becomes poor.

Key words: LaCoO<sub>3</sub>, PrCoO<sub>3</sub>, DyCoO<sub>3</sub>, synthesis, mechanochemistry, crystallization

### **1. INTRODUCTION**

Perovskite lanthanoid transition-metal oxides are candidate as redox catalysts and gas sensors, electrode materials for fuel cells, and so on [1,2]. These materials are synthesized by solid-state reactions and recently by build-up processes such as sol-gel methods. The authors have demonstrated the synthesis of perovskite-LnFeO<sub>3</sub> (Ln: La. Pr. Sm. Gd, Dy, and Yb) by a mechanochemical route in previous papers [3,4]. In this synthesis process, the starting Ln<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>O<sub>3</sub> powders have been ground in organic liquids containing a small amount of  $H_2O$ , where  $Ln_2O_3$  provides a  $Ln(OH)_3$ colloids acting as a reactive grinding aid. The other starting material, Fe<sub>2</sub>O<sub>3</sub>, is kneaded into the colloid to form complex oxyhydroxides. Slightly hydrated  $LnFeO_3 \cdot xH_2O$  (x=0.9-1.4) has been obtained by the dehydration-condensation of the intermediate oxyhydroxide under the grinding stress.

In the present paper, this mechanochemical process has been applied to the synthesis of  $LnCoO_3$  (Ln: La, Pr, Dy). The reactivity is discussed in relation to the potential of the starting materials to redox and hydration reactions in the combination of  $Co^{II}Co^{III}_2O_4$  or  $Co^{II}(OH)_2$  and  $La_2O_3$ ,  $Pr_6O_{11}$  or  $Dy_2O_3$ . The effect of  $H_2O_2$  as an oxidant in the grinding liquid has also been studied.

### 2. EXPERIMENATL

High-purity lanthanoid oxide powders (purity: >99.9%) supplied from Hokko Chem. Ind. were used after the heat-treatment at 500°C for 3h. The specific surface area (SSA) was 6.5 (La<sub>2</sub>O<sub>3</sub>), 2.2 ( $Pr_6O_{11}$ ), and 1.6 m<sup>2</sup>·g<sup>-1</sup> (Dy<sub>2</sub>O<sub>3</sub>). The cobalt source

used, Co<sub>3</sub>O<sub>4</sub> and Co(OH)<sub>2</sub>, was the reagent-grade powders (Kanto-Kagaku). Mechanical grinding was conducted by the use of a planetary mill (Kurimoto-Tekko-Sho). Weighed mixture of the starting powders (20g as LnCoO<sub>3</sub>) was encapsulated into the grinding vessel (capacity: 410cm<sup>3</sup>) made of stainless-steel with acetone (76cm<sup>3</sup>), a small portion of H<sub>2</sub>O (equivalent amount to LnCoO<sub>3</sub> · 4H<sub>2</sub>O composition including  $H_2O$  for  $Co(OH)_2$ ), and grinding media (\$\$\phi2, YTZ balls, fractional filling: 0.40). Grinding was performed for 3h with 15min cooling interval in every 1h to maintain the temperature below 50°C. The ground products were filtered with 0.2µm Teflon membrane, washed with acetone, and dried at 85°C under vacuum. The ground products were calcined at 300-1000°C for 1h in air for crystallization.

Crystalline phases were identified by X-ray diffractometry (XRD, Cu $K\alpha_1$ , 40kV, 20mA) and X-ray photoelectron spectroscopy (XPS, Mg $K\alpha$ , 8kV, 30mA). Dehydration behavior of the ground products was examined by TG-DTA (10 K · min<sup>-1</sup>). Morphology was observed by scanning electron microscopy (SEM, Hitachi, S4300). SSA was determined by a N<sub>2</sub>-adsorption BET method.

# 3. RESULTS AND DISCUSSION 3.1 TG-DTA of the starting Co(OH)<sub>2</sub>

Figure 1 shows the thermal decomposition behavior of  $Co(OH)_2$ . The thermogravimetric curve indicated the mass loss at 200°C, 210-430°C and 930°C. The initial decomposition process at 200°C was the dehydration associated with the oxidation of  $Co^{II}$ . An exothermic peak was observed at 204°C



and the determined mass loss  $(\Delta W_{obs} = 10.1_9\%)$  agreed with the calculation, 10.78% on reaction (1).

$$2Co^{II}(OH)_2 + 1/2O_2 \rightarrow Co^{III}_2O_3 + 2H_2O$$
 (1)

The  $Co_2O_3$  formed was instable and decomposed to  $Co_3O_4$  at 210-430°C. The  $\Delta W_{obs}$ , 13.64%, was almost equal to 13.90% for reaction (2).

$$3Co^{II}(OH)_2 + 1/2O_2 \rightarrow Co^{II}Co^{III}_2O_4 + 3H_2O$$
 (2)

 $Co_3O_4$  further decomposed to CoO above 930°C. This observation was referred to estimate the composition of the ground products.

# 3.2 Bond characteristics of the ground products

The XPS spectra of the ground products are shown in **Fig. 2**. For LC1 using  $Co_3O_4$ , the La3d peaks (844.2 and 848.1eV) and O1s peak (534.2eV) were very close to those of La(OH)<sub>3</sub>, and the Co2P peak was negligibly weak. This indicated the less interaction of  $Co_3O_4$  and La(OH)<sub>3</sub> and the surface of  $Co_3O_4$  covered by La(OH)<sub>3</sub>. It was estimated that the pulverized  $Co_3O_4$  was knead with the adhesive La(OH)<sub>3</sub>. The La3d peak for LC2 and



Fig. 2 XPS spectra of ground products.

LC3 appeared at 845.2 and 842.0eV, which were a little higher energy than those for LaCoO<sub>3</sub>. The Co3d peak for LC2 and LC3 (782.1 eV) was close to that of LaCoO<sub>3</sub> rather than Co(OH)<sub>2</sub>. The O1s spectra also showed the  $E_B$  (531.7e for LC2, 532.2eV for LC3) close to LaCoO<sub>3</sub>, Co(OH)<sub>3</sub> and La(OH)<sub>2</sub> However, the shoulder of O1s spectrum for LaCoO<sub>3</sub> was not observed. Then, the hydrated LaCoO<sub>3</sub>, La,Co-hydroxide, or La,Co-oxyhydroxide were proposed as the structure of the ground products LC2 and LC3. The progress in reactions was suggested for LC3 because of the closer  $E_B$  of O1s peak to LaCoO<sub>3</sub>.

The La3d and O1s XPS spectra for LC1 calcined at 400 and 700°C unchanged from those of the ground product, and the Co2p peak started to appeare at 400°C and increased at 700°C (Fig. 3). The La3d and Co2p spectra for LC2 and LC3 at 300°C unchanged from those of their ground products, but the O1s spectrum for LC3 showed the shoulder similar to LaCoO<sub>3</sub> at  $E_B = 530$ eV. LaCoO<sub>3</sub> should form at the quite low-temperature of 300°C. At 1000°C, all the ground products showed the XPS spectra for LaCoO<sub>3</sub>.

### 3.3 Crystallization of the ground products

The estimation from XPS was ascertained by XRD. Figure 4 shows the XRD patterns of the ground and calcined LC1. As estimated from the XPS results, the ground product was the mixture of La(OH)<sub>3</sub> and Co<sub>3</sub>O<sub>4</sub>. La(OH)<sub>3</sub> was dehydrated and crystallized to La<sub>2</sub>O<sub>3</sub> at 600-700°C. The reflections of Co<sub>3</sub>O<sub>4</sub> remained upto 700°C. However, the reflections for Co<sub>3</sub>O<sub>4</sub> were evidently small against those of the mortar-mixed La(OH)<sub>3</sub>-1/3 Co<sub>3</sub>O<sub>4</sub>, meaning the formation of the mixed hydroxide in part. The yield ( $\Psi$ ) of the ground product was evaluated to be 43% by comparing the peak intensity of La(OH)<sub>3</sub> in LC1 to that of the mortar-mixed one. The ground products LC2 and



Fig. 3 Dependence of XPS spectra on calcination temp.



Fig. 4 XRD profiles of the ground products and calcined powders;  $\bigcirc$ : LaCoO<sub>3</sub>,  $\diamondsuit$ : Co<sub>3</sub>O<sub>4</sub>,  $\checkmark$ : La(OH)<sub>3</sub>,  $\triangle$ : La<sub>2</sub>O<sub>3</sub>.

LC3 contains small amounts of La(OH)<sub>3</sub>. The yield ( $\Psi$ ) for LC2 and LC3 increased to 87% and 88%, respectively. The ground products were amorphous at 300-500°C, crystallized to pseudo-tetragonal LaCoO<sub>3</sub> at 600°C, and converted to rhombohedral phase above 800°C without forming any by-product. These observations were consistent with the discussion on XPS spectra.

# 3.4 TG-DTA of the ground products

Figure 5 shows the TG-DTA curves of LC1, LC2 and LC3. The ground product LC1 showed the mass loss at 400 and 700°C. The observed mass loss ( $\Delta W_{obs}$ ) at 100-1000°C was 10.3<sub>6</sub>%, which agreed with 9.01% for the following reaction (3) when considering a little hydration of Co<sub>3</sub>O<sub>4</sub>. The ground product LC2 and LC3 indicated  $\Delta W_{obs}$  of 13.3<sub>4</sub>% and 12.8<sub>6</sub>% at 1000°C, respectively. These values agreed with  $\Delta W_{cale}$ , 13.09% and 12.78%, for the reactions (4) and (5).

$$La(OH)_3 + 1/3Co_3O_4 + 1/12O_2 \rightarrow LaCoO_3 + 3/2H_2O$$
 (3)

$$LaCo(OH)_5 + 1/4O_2 \rightarrow LaCoO_3 + 5/2H_2O$$
 (4)



Fig. 5 TG-DTA curves of LC1, LC2 and LC3.

$$LaCoO(OH)_4 \rightarrow LaCoO_3 + 2H_2O$$
 (5)

The DTA curve for LC2 indicated the exothermic peak for the oxidation of Co<sup>II</sup> to Co<sup>III</sup> at 170°C. However it was not observed for LC3. The hydrogen peroxide doped into the grinding liquid oxidized Co<sup>2+</sup> to Co<sup>3+</sup> with low hydration affinity, then, leads the formation of oxyhydroxide ( $\Delta W_{calc}$ =12.78%) rather than hydroxide. The LaCo(OH)<sub>5</sub> (LC2) and LaCoO(OH)<sub>4</sub> (LC3) formed amorphous LaCoO<sub>3</sub> at 400-600°C and crystallized at 641°C (LC2) and 609 °C (LC3) with about 2% mass loss.

#### 3.5 Synthesis of PrCoO<sub>3</sub> and DyCoO<sub>3</sub>

This mechanochemical process was applied to the synthesis of  $PrCoO_3$  and  $DyCoO_3$ . Figure 6 shows the XRD patterns of the ground product and the calcined powders for  $Pr_6O_{11}$ -Co(OH)<sub>2</sub> system without doping  $H_2O_2$ . On the synthesis of praseodymium ferrite  $PrFeO_3$ ,  $Pr_6O_{11}$ , shows poor reactivity with FeO(OH) under grinding, and the XRD reflections of  $Pr_6O_{11}$  remained even at 700°C [3]. It was considered that the high stability



Fig. 6 XRD profiles for  $Pr_6O_{11}$ -Co(OH)<sub>2</sub> system.

as oxide, that is the low affinity to hydration, of  $Pr^{IV}$  was the reason for the poor reactivity. So, when the Co<sup>II</sup> can reduce the  $Pr^{IV}$  to  $Pr^{III}$ , there is a possibility to progress the formation of hydroxide or oxyhydroxide. The XRD patters of the ground product PC indicated only the small reflections of  $Pr(OH)_3$  and no reflections from  $Pr_6O_{11}$ . The mass loss value ( $\Delta W_{obs}$ =12.17%) was comparable to the calculated value ( $\Delta W_{calc}$ =12.69%) for the oxyhydroxide,  $PrCOO(OH)_4$ . The yield of  $PrCOO(OH)_4$  in the ground product ( $\Psi$ ) was obtained to be 88% from the peak intensity of  $Pr(OH)_3$ . The overall reaction can be represented as the following equation.

$$1/6Pr^{III}_{2}Pr^{IV}_{4}O_{11} + Co^{II}(OH)_{2} + H_{2}O + 1/12O_{2}$$
  
 $\rightarrow Pr^{III}Co^{III}O(OH)_{4} (6)$ 

The small amount of  $O_2$  required in (6) would be supplied from the ambient atmosphere under grinding or calcination. Tetragonal  $PrCoO_3$  was successfully synthesized above  $600^{\circ}C$ .

Another example of the application was the synthesis of  $DyCoO_3$ . The dysprosium ferrite,  $DyFeO_3$ , is one of the most easily synthesized materials in the series of  $LnFeO_3$ . However, contrary to the  $Pr_6O_{11}$ - $Co(OH)_2$  system, the reactivity in  $Dy_2O_3$ - $Co(OH)_2$  system markedly degraded. The reflections for  $Dy_2O_3$  were observed in the grinding product and their intensity increased with the raise in temperature. The reason for the poor reactivity of DC was unclear; however, the addition of the oxidant should be needed.

#### 3.6 Morphology of ground and calcined LC2, PC, DC

The SSA and morphology of the ground and calcined powders are sited in **Table 1** and **Fig. 8**. The SSA of the ground products decreased with the increase in reactivity except DC. The SEM photographs in Fig. 8 showed the agglomeration. This was strongly related to the mechanism expected, where the adhesive  $Ln(OH)_3$  colloids were used as the reactive grinding aid.



Fig. 7 XRD profiles for Dy<sub>6</sub>O<sub>3</sub>-Co(OH)<sub>2</sub> system.

Table 1.	Specific surface area (SSA) of the ground
	and calcined nowders

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	Ground	round Calcined powder			er		
		400°C	700°C	800°C	1000°C		
LC1	47.5	42.8	6.9	5.3	1.000		
LC2	23.9	13.2	3.9		1.4		
LC3	17.9	11.4	6.9	5.1	-		
PC	12.2	10.2	3.0	-	1.7		
DC	11.7	8.7	5.2		1.3		



Fig. 8 Morphology of ground products.

### 4. CONCLUSION

The mechanochemical process is successfully applied to the synthesis of LaCoO<sub>3</sub>. The ground products obtained from La<sub>2</sub>O<sub>3</sub> and Co(OH)<sub>2</sub> are amorphous complex hydroxide and converted to prseudo-tetragonal LaCoO<sub>3</sub> at 600°C and rhombohedral one above 800°C. The addition of H<sub>2</sub>O<sub>2</sub> to the grinding liquid results in the oxidation of Co<sup>II</sup> to Co<sup>III</sup> to contribute the formation of oxyhydroxide, LaCoO(OH)<sub>4</sub>, which can be dehydrated to amorphous LaCoO<sub>3</sub> at 300°C. The reduction activity of Co<sup>II</sup>(OH)<sub>2</sub> enables the formation of PrCoO<sub>3</sub> from Pr<sub>6</sub>O<sub>11</sub>, but makes the synthesis of DyCoO<sub>3</sub> difficult.

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(Recieved :January 15, 2008; Accepted June 20, 2008)